

1 **Effect of hydrogen-bond strength on photoresponsive properties of polymer-**  
2 **azobenzene complexes**

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17

18 **Abstract:** Supramolecular complexation between photoresponsive azobenzene chromophores and  
19 a photopassive polymer host offers synthetic and design advantages compared to conventional  
20 covalent azo-containing polymers. In this context, it is important to understand the impact of the  
21 strength of the supramolecular interaction on the optical response. Herein, we study the effect of  
22 hydrogen-bonding strength between a photopassive polymer host [poly(4-vinylpyridine), or  
23 P4VP)] and three azobenzene analogues capable of forming weaker (hydroxyl), stronger  
24 (carboxylic acid), or no H-bonding with P4VP. The hydroxyl-functionalized azo forms complete  
25 H-bonding complexation up to equimolar ratio with VP, while the COOH-functionalized azo  
26 reaches only up to 30% H-bond complexation due to competing acid dimerization that leads to  
27 partial phase separation and azo crystallization. We show that the stronger azo-polymer H-bonding  
28 nevertheless provides higher photoinduced orientation and better performance during optical  
29 surface patterning, in terms of grating depth and diffraction efficiency, when phase separation is  
30 either avoided altogether or is limited by using relatively low azo contents. These results  
31 demonstrate the importance of the H-bonding strength on the photoresponse of azopolymer  
32 complexes as well as the need to take into account the interplay between different intermolecular  
33 interactions that can affect complexation.

34

35 **Key words:** supramolecular complexes, azo-containing materials, photo-orientation, surface relief  
36 gratings

37

## 38 **Introduction**

39       Supramolecular chemistry, based on specific noncovalent interactions between  
40 complementary molecules, is an excellent tool for building libraries of functional materials, since  
41 with little effort one structural material element at a time can be varied.<sup>1-5</sup> This strategy has been  
42 utilized, for instance, for establishing structure-property relationships in polymer-azobenzene  
43 complexes.<sup>6-14</sup> These complexes, and azo-containing amorphous materials in general, can convert  
44 incident light energy directly into motion, leading to photoinduced dichroism and birefringence as  
45 well as to the formation of topological surface patterns such as surface-relief gratings (SRGs) by  
46 exposure to interference patterns of light.<sup>15-22</sup>

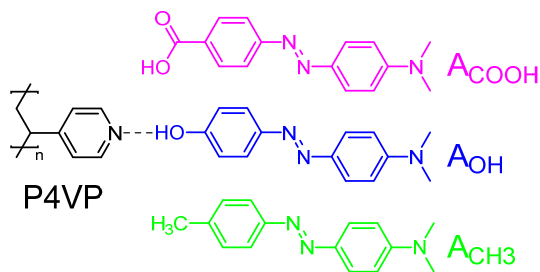
47       Since supramolecular materials design relies on spontaneously forming intermolecular  
48 physical interactions, it is important to understand the effect of the supramolecular interaction  
49 strength on the optical phenomena. To date, there have been relatively few such investigations in  
50 polymer-azobenzene complexes. Notably, for halogen-bonding interactions, the non-linear optical  
51 response of liquid crystals<sup>23</sup> as well as the photo-orientation<sup>24</sup> and all-optical surface patterning  
52 efficiency<sup>25</sup> have been observed to correlate with halogen-bonding strength. Similarly, a  
53 comparison between analogous material systems consisting of hydrogen bonding, ionic bonding  
54 and mixed hydrogen and ionic bonding<sup>26</sup> demonstrated a dependence of the optical surface  
55 patterning efficiency on the type of supramolecular bonding, and thus presumably their strength.

56       As a natural corollary and given that hydrogen (H) bonding is the most commonly used  
57 supramolecular interaction, it is of interest to examine if there is a similar relationship between H-  
58 bond strength and optical responses in H-bonded azomaterials. Thus, in this study, we investigate  
59 a small library consisting of poly(4-vinylpyridine) (P4VP) as the H-bond acceptor polymer host  
60 and two H-bond donor chromophores ( $A_{\text{COOH}}$  and  $A_{\text{OH}}$ , A denoting the azo derivative and COOH

61 and OH the H-bond donating head groups) over a range of  $A_{\text{head}}:\text{VP}$  molar ratios ( $x = 0.05-1.0$ ),  
62 where the H-bonds of  $A_{\text{COOH}}$  and  $A_{\text{OH}}$  with P4VP are considered to be strong and medium-strength,  
63 respectively.<sup>27,28</sup> For comparison, a chromophore with a methyl head group ( $A_{\text{CH}_3}$ ), which has no  
64 specific functional group interactions with P4VP, is also studied. The molecular structures and  
65 nomenclature of the components are given in Scheme 1. Infrared spectroscopy, UV-visible  
66 spectroscopy and polarized optical microscopy are used to characterize the degree of complexation  
67 and optical properties of the materials. We then compare the photoresponse of the three systems  
68 in terms of photo-orientation upon illumination with linearly polarized light and of SRG depth and  
69 diffraction efficiency upon irradiation with an interference pattern of light.

70

71 **Scheme 1.** Chemical structures and nomenclature of the polymer and chromophores used.



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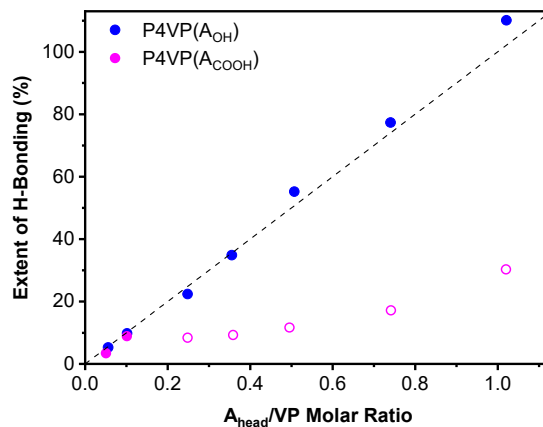
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## 74 Results and discussion

75 First, the extent of H-bonded pyridine units in the  $\text{P4VP}(A_{\text{OH}})_x$  and  $\text{P4VP}(A_{\text{COOH}})_x$   
76 complexes [not  $\text{P4VP}(A_{\text{CH}_3})_x$  since these have no H-bonding interactions] as a function of azo  
77 content was determined by infrared spectroscopy following the methodology established in our  
78 previous work.<sup>8</sup> The results in Fig. 1 demonstrate that the degree of H-bonding in the  $\text{P4VP}(A_{\text{OH}})_x$   
79 complexes reaches the nominal values indicated by the  $A_{\text{head}}:\text{VP}$  molar ratios used in the sample  
80 preparation, such that a claim of essentially complete H-bonding between  $A_{\text{OH}}$  and P4VP can be  
81 made. On the other hand, the  $\text{P4VP}(A_{\text{COOH}})_x$  complexes depart from this trend already at  $x = 0.25$ ,

82 where the measured degree of H-bonding is less than half the nominal value and never exceeds  
83 30% up to the nominal equimolar ratio. The low degree of H-bonding with P4VP can be related to  
84 competing interactions, notably among the COOH groups, which are highly subject to  
85 dimerization.<sup>29,30</sup> As will be shown below, this incomplete H-bonding favours phase separation in  
86 the P4VP(A<sub>COOH</sub>)<sub>x</sub> system and thus affects the optical clarity of the thin films. Nevertheless,  
87 despite this difference in the degree of H-bonding between the two chromophores with P4VP for  
88  $x = 0.25$  and above, it is possible to compare the effect of the H-bond strength on photo-orientation  
89 and SRG inscription for at least  $x = 0.05$  and  $0.10$ , for which H-bonding with P4VP is complete in  
90 both sets of complexes, keeping in mind also that it has been shown that A<sub>OH</sub> concentrations as  
91 low as  $x=0.01$  are enough for photoinduced surface patterning.<sup>31</sup> In addition, it is of interest to  
92 examine, for comparison, the consequences of partial H-bonding on the optical properties at high  
93 degrees of complexation. It is also worth mentioning here that spin-coating, which allows rapid  
94 solvent removal, and drying at room temperature, are necessary to avoid even greater phase  
95 separation in the P4VP(A<sub>COOH</sub>)<sub>x</sub> complexes, as found also for similar complexes with another  
96 polymer.<sup>26</sup>

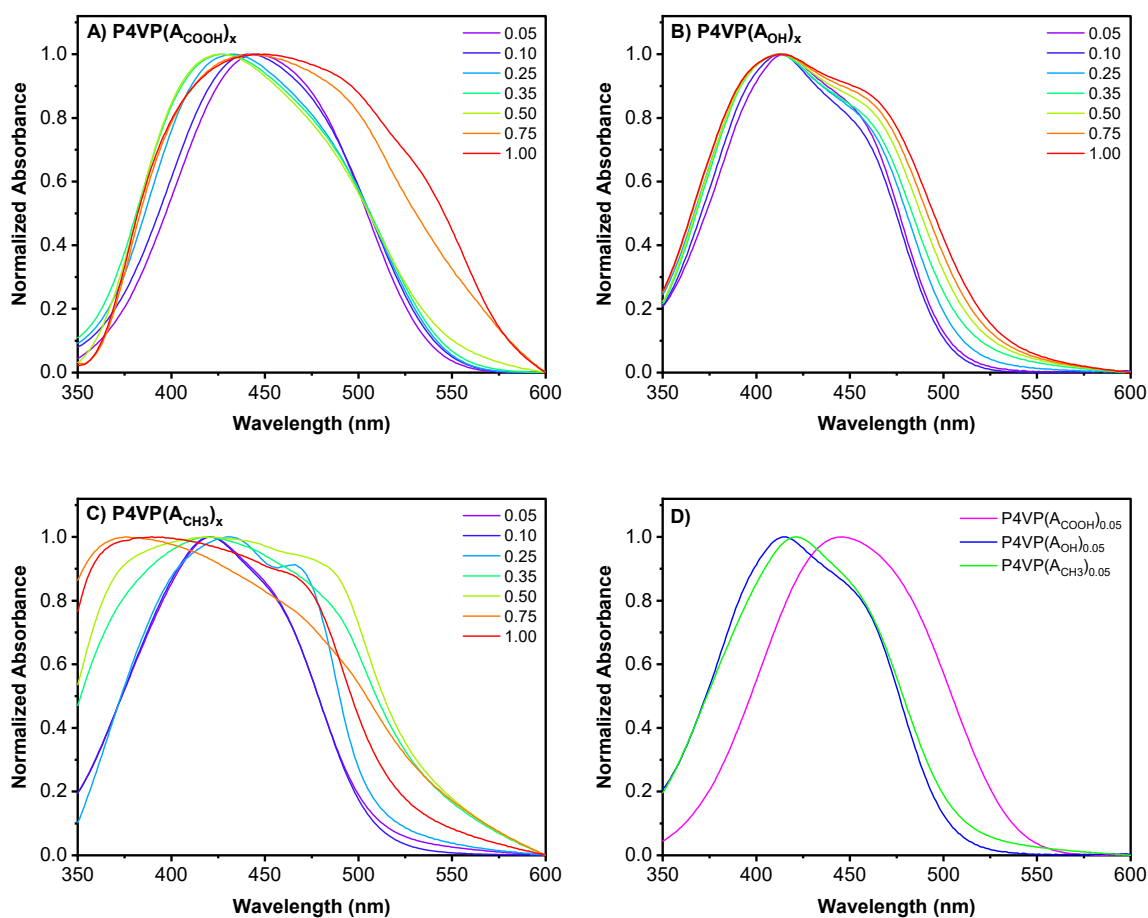
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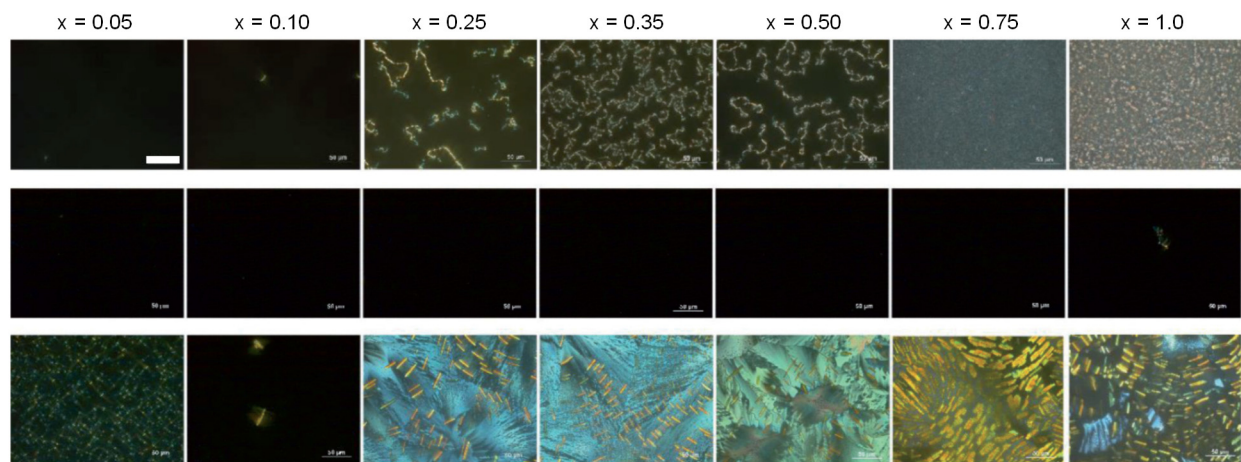
98  
 99 **Fig. 1.** Degree of hydrogen bonding in the P4VP(AOH)<sub>x</sub> and P4VP(ACOOH)<sub>x</sub> complexes. The dashed  
 100 line represents complete complexation. Closed and open symbols indicate H-bonding of A<sub>head</sub> to  
 101 P4VP that is complete and partial, respectively.

102  
 103 The UV-visible spectra of the P4VP(AOH)<sub>x</sub> system do not show a significant shift of the  
 104 absorption maximum over the complete range of x, confirming that the weakly interacting A<sub>OH</sub>  
 105 chromophores remain unaggregated up to equimolar degrees of complexation (Fig. 2B). This is  
 106 consistent with the chromophores being completely H-bonded to the polymer, as noted in Fig. 1.  
 107 In contrast, for both P4VP(ACOOH)<sub>x</sub> and P4VP(A<sub>CH3</sub>)<sub>x</sub>, chromophore-chromophore interactions  
 108 manifest themselves at and above x = 0.25 (Fig. 2A and 2C, respectively), indicative of  
 109 chromophore crystallization or phase separation. For P4VP(ACOOH)<sub>0.25-0.50</sub>, the most obvious effect  
 110 of these interactions is the blue shift of the spectral maxima indicating excitonic coupling (H-  
 111 aggregation) of two or more adjacent chromophores, suggesting small crystals. For  
 112 P4VP(ACOOH)<sub>0.75-1.0</sub>, the spectra are dominated by increased scattering, which can be attributed to  
 113 macrophase separation of larger crystalline chromophore domains. This type of scattering is  
 114 evident also in the spectra of the P4VP(A<sub>CH3</sub>)<sub>x</sub> complexes over the range of x = 0.25-1.0. At lower  
 115 x, it is plausible that most A<sub>CH3</sub> molecules are dispersed in the P4VP despite the absence of specific

116 interactions between  $A_{CH_3}$  and P4VP, as previously observed in mixtures containing up to  $x = 0.08$   
117 of the Disperse Red 1 dye in a non-interacting polystyrene host.<sup>32</sup> Indeed, Fig. 2D shows that for  
118 a low  $x$  value of 0.05, the spectrum of P4VP( $A_{CH_3}$ ) is very similar to that of P4VP( $A_{OH}$ ), consistent  
119 with the similar calculated dipole moments (3.5 and 3.8 D, respectively) of the two chromophores.  
120 By contrast, the maximum absorbance in the spectrum of P4VP( $A_{COOH}$ )<sub>0.05</sub> is red-shifted by  $\sim 25$   
121 nm, consistent with the higher dipole moment of 7.7 D calculated for  $A_{COOH}$ .  
122



123  
124  
125 **Fig. 2.** Normalized UV-visible absorption spectra of thin films of A) P4VP( $A_{COOH}$ )<sub>x</sub>, B)  
126 P4VP( $A_{OH}$ )<sub>x</sub>, and C) P4VP( $A_{CH_3}$ )<sub>x</sub> for the various  $A_{head}/VP$  ratios ( $x$ ) indicated. D) Comparison of  
127 the normalized UV-visible absorption spectra of the three systems at  $x = 0.05$ .



129

130 **Fig. 3.** Polarized optical microscopy images for P4VP(ACOOH)<sub>x</sub> (uppermost line), P4VP(AOH)<sub>x</sub>  
 131 (middle line), and P4VP(ACH<sub>3</sub>)<sub>x</sub> (bottom line). The scale bar (illustrated in the top left image) is 50  
 132 μm for all images.

133

134 Additional evidence of crystallization or lack thereof can be observed in the polarized optical  
 135 microscopy (POM) images of the three systems, illustrated in Fig. 3. The P4VP(AOH)<sub>x</sub> samples are  
 136 non-birefringent – i.e. isotropic – over the whole complexation range investigated, indicating that  
 137 there is neither phase separation with crystallization of the chromophore (in line with the measured  
 138 degrees of complexation and the UV-visible spectra) nor liquid crystal character in the complexes,  
 139 as found also in Ref 33. In contrast, birefringent domains appear in most of the P4VP(ACOOH)<sub>x</sub> and  
 140 P4VP(ACH<sub>3</sub>)<sub>x</sub> samples, which can be related to azo crystallization, in line with the extent of  
 141 complexation and the UV-visible spectra. The P4VP(CH<sub>3</sub>)<sub>x</sub> samples show highly intense and  
 142 essentially continuous birefringence for  $x = 0.25$  and above. This is consistent with what might be  
 143 expected for a system with no specific interactions between the chromophore and the polymer,  
 144 such that extensive and relatively unimpeded crystallization takes place above a low solubility  
 145 limit that depends on the chemical nature of the azo dye and the polymer matrix. For



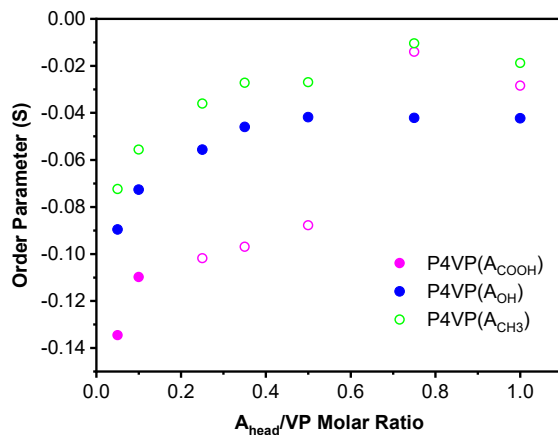
146 P4VP(A<sub>COOH</sub>)<sub>x</sub>, birefringence appears at  $x = 0.25$  in the form of mainly isolated domains whose  
147 density tends to increase with increasing  $x$ . In this case, the H-bond interactions of A<sub>COOH</sub> with  
148 P4VP, even if at a relatively low level, likely impede azo crystallization and limit crystallite size,  
149 thus diminishing the resultant birefringence in POM. It may also be noted that there appears to be  
150 a correlation between the blue-shifted UV-visible spectra of P4VP(A<sub>COOH</sub>)<sub>0.25-0.50</sub> and the POM  
151 images showing isolated birefringent areas on a dark background, indicating the presence of both  
152 optically anisotropic and isotropic domains in these samples, whereas the dominance of scattering  
153 effects in the UV-visible spectra of P4VP(A<sub>COOH</sub>)<sub>0.75-1.0</sub> can be related to the greater density of  
154 birefringent structures in the POM images.

155 For what follows, a word should be said about the glass transition temperature ( $T_g$ ). The  $T_g$   
156 of the P4VP used is about 125 °C.<sup>34</sup> The addition of H-bonded azo derivatives generally decreases  
157 the  $T_g$  with added azo due to a plasticizing effect which can be followed by an increase in  $T_g$  as  
158 full complexation is approached due to decreasing mobility of the highly complexed azos, such  
159 that the minimum  $T_g$  reached is roughly 60 °C for azos similar to those used here.<sup>6,26,33-34</sup>  
160 Furthermore, in the presence of phase-separated (crystallized) azo, the P4VP  $T_g$  is less affected.<sup>26</sup>  
161 We conclude that, because the system  $T_g$  is always well above ambient temperature, it is unlikely  
162 that the precise  $T_g$ s of the systems under study have a large effect on the optical properties  
163 described below.

164 Fig. 4 illustrates the order parameter of the chromophores that is reached in the complexes  
165 after illumination with linearly polarized 488 nm light. This pump wavelength, which was also  
166 used for the SRG inscription experiments, is absorbed by both the trans ( $\pi - \pi^*$  band) and cis ( $n -$   
167  $\pi^*$  band) conformers of all three azobenzene compounds studied. This promotes rapid trans–cis  
168 and cis–trans photochemical reactions as is needed for efficient photo-orientation and all-optical

169 SRG inscription. This light orients the chromophores in the direction perpendicular to the incident  
170 polarization, hence giving negative S values. The maximum orientation is obtained at low x for all  
171 samples and decreases with increasing x. A similar decrease in azo orientation with increasing x  
172 was found for other H-bonded<sup>8</sup> and halogen-bonded<sup>24</sup> azopolymer complexes. However, the trend  
173 in the decrease of S is smooth for P4VP(A<sub>OH</sub>) complexes over the whole range of x and tends  
174 toward a plateau, analogously to what was observed for saturated photoinduced birefringence  
175 normalized by number density of chromophores in this material.<sup>33</sup> In contrast, an abrupt change is  
176 observed for P4VP(A<sub>COOH</sub>) above x=0.5, and possibly also for P4VP(A<sub>CH<sub>3</sub></sub>) (although the already  
177 low S at x = 0.35-0.5 makes this difficult to see clearly), both of which contain crystallized  
178 chromophore at higher x.

179



180

181 **Fig. 4.** Order parameter after illumination of 300 s with linearly polarized 488 nm light. Open  
182 symbols indicate partial crystallization of the azobenzene chromophores.

183

184 At low x, where aggregation is minimal or inexistent, the P4VP(A<sub>COOH</sub>) system reaches a  
185 substantially higher orientation than P4VP(A<sub>OH</sub>) and P4VP(A<sub>CH<sub>3</sub></sub>). While this partially reflects a  
186 tendency of azo compounds with a higher dipole moment to orient more due to more efficient

187 trans-cis-trans isomerization cycling,<sup>35,36</sup> differences in absorption coefficients of the three  
188 chromophores at 488 nm (see Fig. 2D) may also contribute to the higher order parameters reached  
189 by A<sub>COOH</sub>. The detrimental effect of chromophore aggregation on photo-orientation is clearly  
190 visible in this data, since the orientation values for P4VP(A<sub>COOH</sub>)<sub>0.75-1.0</sub> become lower than for  
191 P4VP(A<sub>OH</sub>)<sub>0.75-1.0</sub> which forms weaker H-bonds but resists phase separation. Values for  
192 P4VP(A<sub>COOH</sub>)<sub>0.75-1.0</sub> reach the same low orientation level as those for P4VP(A<sub>CH3</sub>)<sub>0.75-1.0</sub> having no  
193 specific supramolecular interaction with the polymer backbone and thus certainly subject to  
194 chromophore-chromophore intermolecular interactions.

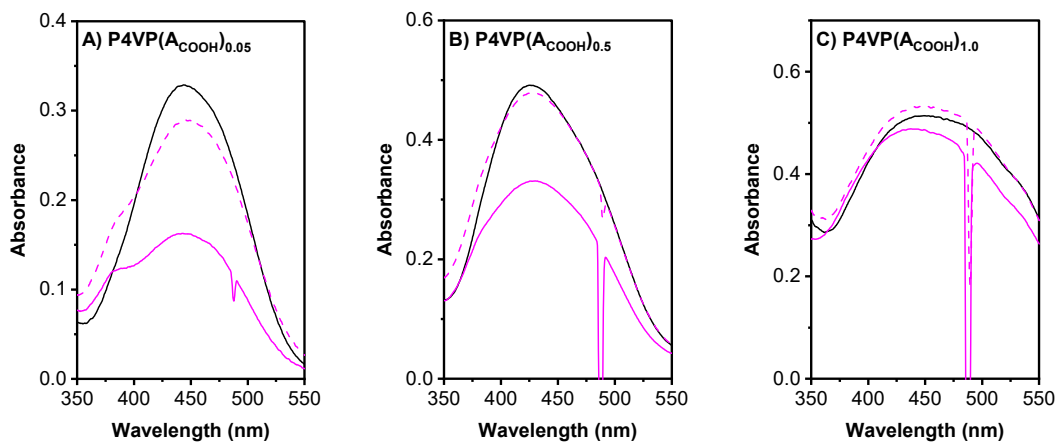
195       Generally, the values of S do not distinguish between whether the photo-orientation is caused  
196 by angular hole burning (AHB) due to polarization-selective azobenzene trans-cis isomerization,  
197 by angular redistribution (AR) of the long axis of the trans molecules, or by both.<sup>37,38</sup> More insight  
198 can be gained by comparing, as done in Fig. 5, the polarized absorption spectra under irradiation  
199 with the spectrum measured in the dark before irradiation. As expected, the parallel-polarized  
200 absorbance (full lines) decreases upon irradiation as both AHB and AR processes reduce the  
201 number of trans conformers in the direction parallel to the laser polarization. The behaviour of the  
202 perpendicular-polarized spectra (dashed lines) is more distinctive. With  $x = 0.05$ , the perpendicular  
203 absorbance after 5 min of irradiation is lower than in the original spectrum, and not higher as one  
204 would expect if the azobenzene chromophores were simply reoriented under irradiation. A biaxial  
205 orientation<sup>39</sup> may partially contribute to this observation but it cannot be the dominating effect  
206 because it would imply a very strong tendency to orient in the out-of-plane direction, which is  
207 extremely unlikely for an amorphous well-dispersed system. We thus interpret the decreasing  
208 perpendicular-polarized absorbance as evidence that angular hole burning is the dominant effect  
209 responsible for the S for all three systems (panels A, D and G of Fig. 5) at low x. In contrast, at x

210 = 1.0, the absorbance for P4VP(A<sub>COOH</sub>) becomes larger than the initial value, a clear indication of  
211 angular redistribution (Fig. 5C). The relative changes in parallel and perpendicular absorbances  
212 compared to the initial spectrum in the dark suggest that the uniaxial model is an appropriate  
213 approximation to calculate the S values. After 5 min of relaxation in the dark (not shown), the  
214 perpendicular-polarized absorbance further increases due to the reversed AHB process, the cis-  
215 trans isomerization. The relative importance of the AR mechanism appears to gradually increase  
216 for P4VP(A<sub>COOH</sub>)<sub>x</sub> systems of increasing x value, as illustrated in Fig. 5B for P4VP(A<sub>COOH</sub>)<sub>0.5</sub>. A  
217 similar trend can be noted for P4VP(A<sub>OH</sub>)<sub>x</sub>, although the absorbance with perpendicular  
218 polarization does not exceed the dark spectrum even for P4VP(A<sub>OH</sub>)<sub>1.0</sub>. Because of strong  
219 scattering, it is not possible to firmly establish the behaviour of P4VP(A<sub>CH3</sub>)<sub>x</sub> with x = 0.5 or 1.0.  
220 These results support our previous observations that stronger chromophore-chromophore  
221 interactions promote the angular redistribution mechanism and that its relative importance,  
222 compared to AHB, increases at higher azobenzene content.<sup>8</sup> In view of this, the partial aggregation  
223 of the chromophores could improve the photoresponse of the system, as it does for liquid-  
224 crystalline azopolymers,<sup>7,14</sup> to the extent that the optical properties of the thin films remain  
225 acceptable.

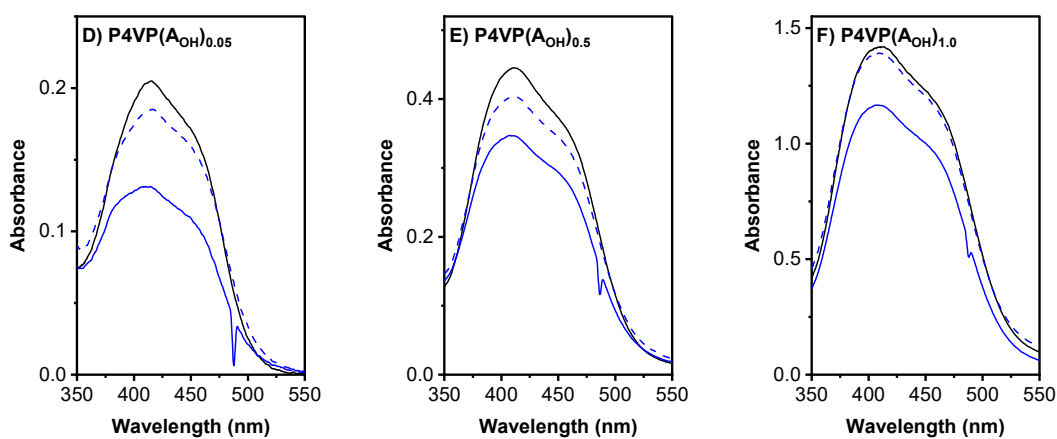
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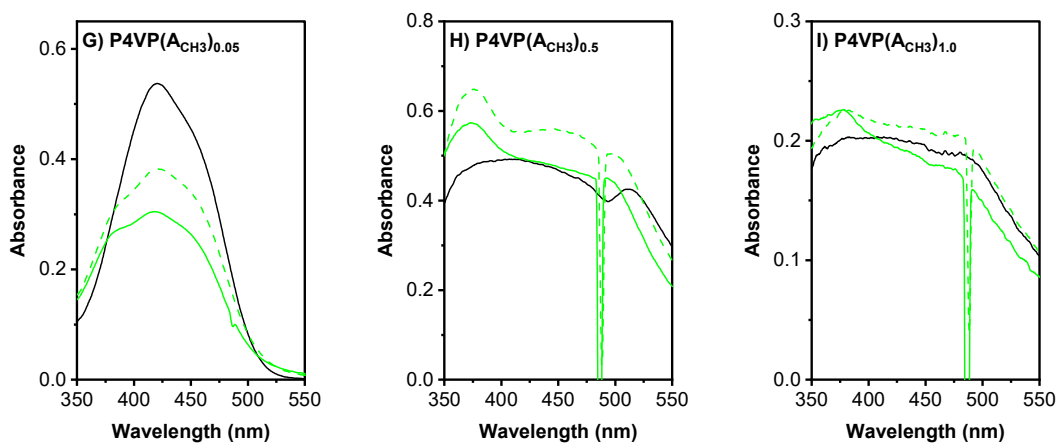
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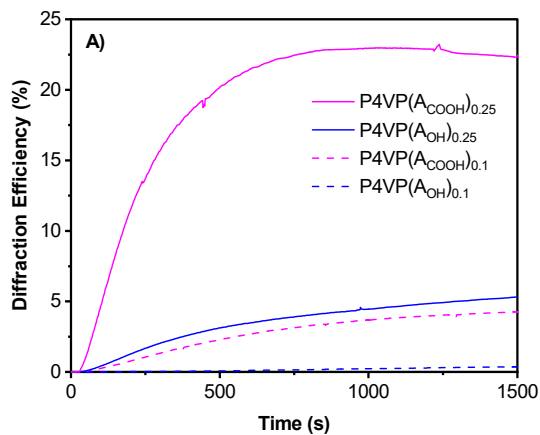


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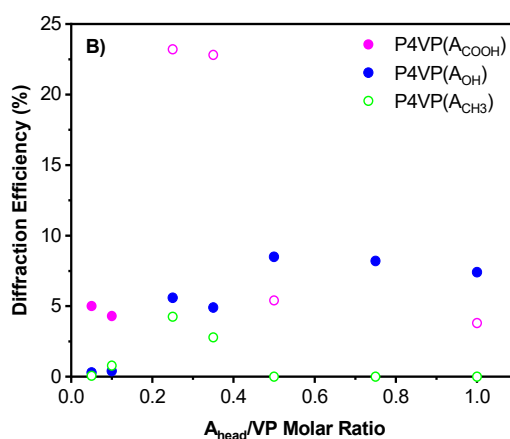


231 **Fig. 5.** Comparison of the UV-visible spectra polarized parallel and perpendicular to the laser  
 232 polarization direction (full and dashed lines, respectively) recorded after 5 min of 488 nm laser  
 233 irradiation. The spectra prior to illumination are shown as black lines. The sharp features at 488  
 234 nm are due to stray laser light.

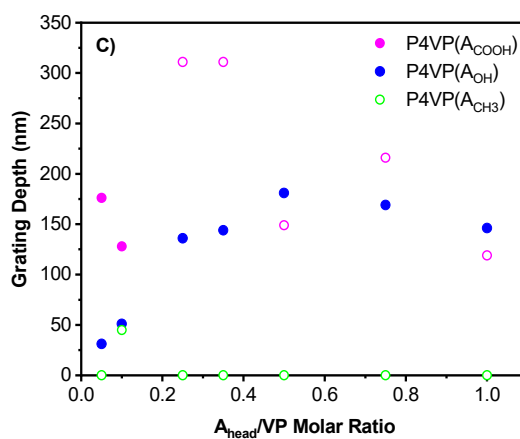
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238 **Fig. 6.** A) Representative SRG inscription curves, B) maximum diffraction efficiency reached  
239 upon SRG inscription, and C) the corresponding grating modulation depth measured by AFM.  
240 Open symbols indicate crystallization of the azobenzene chromophores.

241

242 SRG inscription on thin films of the different samples demonstrates the same interplay of  
243 intermolecular interaction strength and molecular structure, as do the absorption and photo-  
244 orientation results. Figure 6A illustrates the SRG writing dynamics for P4VP(A<sub>OH</sub>) and  
245 P4VP(A<sub>OH</sub>) with  $x = 0.1$ , where the azo is well dispersed in both complexes, and with  $x = 0.25$ ,  
246 where phase separation has already begun for P4VP(A<sub>COOH</sub>). The diffraction efficiency smoothly  
247 increases with time in all cases but at a much faster rate for P4VP(A<sub>COOH</sub>) and with increasing  $x$ .  
248 Qualitatively, saturation of the diffraction efficiency is reached within 1500 s only for the  
249 P4VP(A<sub>COOH</sub>)<sub>0.25</sub> sample. Figure 6B plots the maximum diffraction efficiency reached by all  
250 samples. The most important observation, in terms of our initial objective of comparing the effect  
251 of the H-bond strength, is that at low degrees of complexation ( $x = 0.05-0.10$ ), where there is  
252 maximal H-bonding with P4VP and no A<sub>COOH</sub> phase separation, the diffraction efficiencies  
253 reached are clearly higher for the P4VP(A<sub>COOH</sub>) films than for the P4VP(A<sub>OH</sub>) films. This trend is  
254 mirrored in Figure 6C when plotting the *ex-situ* measured modulation depths of the inscribed  
255 patterns as determined by AFM. This correlation between H-bond strength and SRG response thus  
256 complements the same structure-function relationship observed in halogen-bonded materials<sup>25</sup> and  
257 when comparing H-bonded materials with analogous ionically-bonded and mixed H-  
258 bonded/ionically bonded materials.<sup>26</sup>

259 Even for  $x = 0.25$  and  $0.35$ , where A<sub>COOH</sub> H-bonding to P4VP is drastically reduced and  
260 A<sub>COOH</sub> crystallization has set in (but keeping in mind that part of the non-bonded A<sub>COOH</sub> probably  
261 remains molecularly dispersed within the polymer matrix instead of crystallizing), the  
262 P4VP(A<sub>COOH</sub>) films continue to surpass the P4VP(A<sub>OH</sub>) films by about fourfold and twofold,  
263 respectively, which may be indicative of a relatively low degree of A<sub>COOH</sub> crystallization at these  
264  $x$ .

265 It may be noted that the SRG efficiency and grating amplitude tend to increase with increase  
266 in molar ratio in the lower  $x$  range, up to about 0.35 for P4VP(A<sub>COOH</sub>) and about 0.5 for  
267 P4VP(A<sub>OH</sub>), this much more strongly for the former than the latter as noted in the previous  
268 paragraph. Above  $x = 0.35$ , the response in the P4VP(A<sub>COOH</sub>) films is degraded strongly, which  
269 can be attributed to significant A<sub>COOH</sub> crystallization. In an analogous system, Gao *et al.* also  
270 reported an increase in the SRG modulation depth until only  $x = 0.40$ .<sup>9</sup> Despite the degradation in  
271 response at higher molar ratios, the measured grating depths in the P4VP(A<sub>COOH</sub>) films are similar  
272 to those of the P4VP(A<sub>OH</sub>) films in the 0.45-1.0 range of  $x$ , although the diffraction efficiencies of  
273 the former are a little less. In general, the properties of partially aggregated materials depend on  
274 many parameters of the thin film preparation, for instance on the rate of solvent evaporation, which  
275 particularly influences the degree of phase separation when this tendency is present.<sup>26,30</sup>  
276 Furthermore, the random nature of crystalline nucleation influences final crystalline forms and  
277 sizes.

278 It is noteworthy that the SRG efficiency and amplitude trends for P4VP(A<sub>OH</sub>) <sub>$x$</sub>  as a function  
279 of degree of complexation (Fig. 6) reach their maximum near  $x = 0.5$  and then remain  
280 approximately constant or even decrease a little. This contrasts with what was observed by  
281 Vapaavuori *et al.* for the same complexes,<sup>33</sup> where the response tended to increase up to  $x = 1.0$ .  
282 On the other hand, a similar trend in SRG efficiency and amplitude as a function of  $x$  was observed  
283 in amorphous P4VP-bisazobenzene complexes.<sup>34,40</sup> Some possible explanations are the differences  
284 in the interference patterns used to inscribe gratings, since it is known that the inscription depends  
285 on the initial polarization<sup>41-43</sup> and might also depend on other experimental parameters, such as  
286 film thickness, drying conditions and ambient humidity.



287 Finally, it is of interest to note that there are measurable diffraction efficiencies in the low  
288 molar ratio P4VP(A<sub>CH3</sub>) films, having similar magnitudes to those in P4VP(A<sub>OH</sub>) films for x up to  
289 0.25 but decreasing to 0 for  $x \geq 0.5$ , suggesting that photoisomerization of the chromophores  
290 without specific interactions with the polymer backbone, if (mainly) dispersed in the polymer  
291 matrix, is enough to drive SRG formation. On the other hand, only one sample,  $x = 0.1$ , shows a  
292 measurable SRG amplitude. Further studies, for instance using different diffraction patterns and  
293 benefiting from the recently demonstrated *in-situ* AFM methods,<sup>44,45</sup> are desirable to understand  
294 this phenomenon.

295

## 296 **Conclusion**

297 Varying para-substitution of photoactive azobenzene chromophores is a simple way to alter  
298 their interaction strength with complementary photopassive polymer matrices. This creates a  
299 useful parameter space for studying structure-function relationships of photoinduced motions in  
300 azobenzene-containing supramolecular complexes and for optimizing their photoresponse.  
301 Overall, the above results indicate that for greater SRG and photo-orientation response in H-  
302 bonded azopolymer complexes, it is desirable to use azo derivatives forming stronger H-bonds  
303 with the polymer host, while also maintaining maximal H-bonding and amorphous character up to  
304 high azo molar ratios. COOH-functionalized derivatives outperform their OH-functionalized  
305 analogs when the azo content is sufficiently low to avoid aggregation, since at higher azo content  
306 they tend to crystallize due to acid dimerization, leading to a degradation of the optical  
307 performance. The acid-functionalized system can still outperform the fully complexed hydroxyl-  
308 functionalized system when aggregation is not too pervasive, suggesting that it might be possible  
309 to optimize preparation conditions to give partially aggregated functional materials that perform

310 well. A more promising approach is the molecular design of azobenzene derivatives with a strong  
311 H-bond donating functionality where crystallization is inhibited by, for example, a nearby moiety  
312 that introduces steric hindrance or irregularity. Such system would combine the key characteristics  
313 of maximal complexation and lack of aggregation even at the high azo content needed to produce  
314 supramolecular materials with a strong photoresponse using the common H-bond.

315

## 316 **Experimental**

### 317 **Sample preparation**

318 P4VP ( $M_n = 5100$  g/mol,  $M_w = 5400$  g/mol) was obtained from Polymer Source and the three  
319 azo derivatives from Tokyo Chemical Industry; all were used as received. Mixtures of P4VP and  
320  $A_{\text{head}}$  were prepared by dissolving the constituents individually in DMF, filtering the solutions  
321 through 0.2  $\mu\text{m}$  PTFE syringe filters, and then mixing them at desired  $A_{\text{head}}:\text{VP}$  molar ratios ( $x$ ).  
322 The P4VP( $A_{\text{COOH}}$ ) $_x$  solutions, which tended to precipitate over time, were heated to approximately  
323 90 °C for redissolution just before film preparation. Film samples were prepared by spin-coating  
324 (unless otherwise specified, for 60 s at 800-1200 rpm followed by 15 s at 1500 rpm with a 500  
325 rpm/s acceleration), and left to dry in a fumehood at room temperature overnight or longer. Films  
326 thicknesses generally varied between 100 and 260 nm for P4VP( $A_{\text{COOH}}$ ) and P4VP( $A_{\text{OH}}$ ), and  
327 around 600 nm for P4VP( $A_{\text{CH}_3}$ ).

### 328 **Computations**

329 DFT calculations were done using Gaussian 16 on the supercomputer Graham managed by  
330 Compute Canada. The dipole moment of the azo chromophores were calculated with the B3LYP  
331 functional and the 6-31+G(d,p) basis set. The geometry of the molecules was optimized *in vacuo*.

### 332 **Infrared spectroscopy**

333 Samples for FT-IR characterization were prepared by spin-coating 75  $\mu\text{L}$  solutions of  
334 P4VP( $A_{\text{head}}$ ) in DMF, with a P4VP concentration of 9–10 wt %, on KBr windows at a speed of  
335 700 rpm for 30 s. The samples were dried at room temperature in the container of an FTS Systems  
336 FD-3-85A-MP freeze-dryer equipped with a condenser working at 1–3 mT and  $-90\text{ }^{\circ}\text{C}$  for 2 h  
337 followed by at least 3 d in a covered Petri dish in a fumehood. FT-IR measurements were  
338 performed in transmission on a Bruker Optics Tensor 27 spectrometer with a HgCdTe detector.  
339 All spectra were obtained with a resolution of  $4\text{ cm}^{-1}$  by averaging 512 scans. The fraction of H-  
340 bonded pyridines was determined using the procedures described in Wang *et al.*<sup>8</sup>

#### 341 **UV-visible spectroscopy**

342 Non polarized spectra were recorded with a PerkinElmer Lambda 950 spectrometer. Pure  
343 quartz glass was used as a reference. Photo-orientation studies were performed by illuminating the  
344 thin films with a 488 nm linearly polarized argon laser. The polarized UV-visible spectra were  
345 taken with an Ocean Optics USB2000+ spectrometer before and after 300 s of irradiation. The  
346 light incident to the sample was polarized using a broad-band polarizer, which was manually  
347 rotated between the two in-plane polarizations. The order parameters,  $S$ , were calculated at the  
348 maximum absorbance of each sample using the equation

$$349 \quad S = \frac{A_{\parallel} - A_{\perp}}{A_{\parallel} + 2A_{\perp}}$$

350 where  $A_{\parallel}$  and  $A_{\perp}$  are the absorbances parallel and perpendicular to the laser polarization direction,  
351 respectively.

#### 352 **Polarized optical microscopy**

353 To determine the amorphous or anisotropic (crystalline or liquid crystalline) character of the  
354 samples, polarized optical microscopy images were obtained with a Leica DM4500P polarized  
355 optical microscope.

## 356 **Optical grating formation**

357 The surface-relief grating (SRG) formation was studied by measuring the first-order  
358 diffraction efficiency of a low-intensity 635 nm non-resonant laser diode from the pattern that was  
359 inscribed using a Lloyd's mirror interferometer set-up with an argon laser operating at 488 nm  
360 with approximately 200 mW/cm<sup>2</sup> irradiance. The values reported here as the maximum diffraction  
361 efficiencies were read after 1500 s of inscription. The resulting surface patterns were characterized  
362 by *ex-situ* atomic force microscopy (AFM) measurements carried out using a Veeco Dimension  
363 5000 SPM in tapping mode.

364

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372

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461



462 **Figure captions**

463 **Scheme 1.** Chemical structures and nomenclature of the polymer and chromophores used.

464 **Fig. 1.** Degree of hydrogen bonding in the P4VP(AOH)<sub>x</sub> and P4VP(ACOOH)<sub>x</sub> complexes. The  
465 dashed line represents complete complexation. Closed and open symbols indicate H-bonding of  
466 A<sub>head</sub> to P4VP that is complete and partial, respectively.

467 **Fig. 2.** Normalized UV-visible absorption spectra of thin films of A) P4VP(ACOOH)<sub>x</sub>, B)  
468 P4VP(AOH)<sub>x</sub>, and C) P4VP(ACH<sub>3</sub>)<sub>x</sub> for the various A<sub>head</sub>/VP ratios (x) indicated. D) Comparison of  
469 the normalized UV-visible absorption spectra of the three systems at x = 0.05.

470 **Fig. 3.** Polarized optical microscopy images for P4VP(ACOOH)<sub>x</sub> (uppermost line), P4VP(AOH)<sub>x</sub>  
471 (middle line), and P4VP(ACH<sub>3</sub>)<sub>x</sub> (bottom line). The scale bar (illustrated in the top left image) is 50  
472 μm for all images.

473 **Fig. 4.** Order parameter after illumination of 300 s with linearly polarized 488 nm light. Open  
474 symbols indicate partial crystallization of the azobenzene chromophores.

475 **Fig. 5.** Comparison of the UV-visible spectra polarized parallel and perpendicular to the laser  
476 polarization direction (full and dashed lines, respectively) recorded after 5 min of 488 nm laser  
477 irradiation. The spectra prior to illumination are shown as black lines. The sharp features at 488  
478 nm are due to stray laser light.

479 **Fig. 6.** A) Representative SRG inscription curves, B) maximum diffraction efficiency reached  
480 upon SRG inscription, and C) the corresponding grating modulation depth measured by AFM.  
481 Open symbols indicate crystallization of the azobenzene chromophores.